

Mechanisms Controlling the Degradation of Poly(methyl methacrylate) Prior to Piloted Ignition

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This approach to determining t_{ig} is very useful in evaluation of construction materials for buildings and furniture since it provides a good first order approximation to ranking the materials resistance to ignition as a function of their thermal properties. However, it fails to provide reliable data for the critical thermodynamic and transport properties needed to develop predictive models for ignition and flame spread under conditions different to those in the test. Furthermore, the estimated t_{ig} provides little insight to the ignition process when the environmental conditions are close to the critical conditions for ignition. Good examples of such environments are spacecraft and highly oxygenated environments [4]. Environmental variables become an issue and questions arise about the main hypothesis of the test methodology.

Recent studies have shown that, for many cases, ignition cannot be modeled as the heating of an inert material. The degradation process results in significant gasification of flammable gases before ignition [5]. For example, some materials are fire retardant and their pyrolysis tends to inhibit rather than enhance gas phase combustion [6]. For such class of materials, the period of this mass loss without ignition can last as long or longer than the preheating process to get to T_p . Furthermore, recent studies have noted that a critical mass flux of fuel emerging from the surface [7,8] or a critical fuel concentration in the gas phase are better criteria for predicting ignition than a surface ignition temperature [1,9].

These observations have prompted more complex modeling of the degradation process that question the use of average solid and gas phase properties and have shown that the use of temperature-dependent properties can provide, for certain conditions, better correlation with the experimental ignition delay times. These refinements are reviewed in detail elsewhere [7] and will not be discussed here. More recent efforts include thermal modeling of the pyrolysis vapor generation [1,10], with both decomposition chemistry [11] and in-depth fuel migration (bubbling) [12]. The additional surface/sub-surface models seem necessary to fully describe piloted ignition.

Models that introduce degradation chemistry are more complete and require significant computation and knowledge of the surface chemistry, which is often not available for real materials. Oftentimes, critical steps in the kinetic mechanism remain uncertain and even when they are known, the kinetic constants for some or all of the reaction steps are inadequately defined. Furthermore, a proper model for fuel vapor migration in the sub-surface is also

difficult to define. The lack of data to evaluate detailed surface chemistry and transport mechanisms indicates the need for more fundamental degradation studies before detailed surface chemistry and transport models can be confidently employed for more accurate prediction of piloted ignition. Thus far, the use of global kinetics with experimentally fitted E_a and pre-exponentials is the preferred approach for incorporating degradation chemistry into solid ignition models. The fundamental experimental data on the degradation process will however assess the mechanisms that control the mass flux of fuel evolving from the degrading surface and when models of the various processes can be either neglected or simplified.

The present study uses a thermogravimetric analyzer (TGA) coupled to a quadrupole mass spectrometer (MS) to extract experimental data for a reference material, poly (methyl) methacrylate (PMMA). This data is expected to provide some clarification on the mechanisms controlling the fuel generation in the period that follows the onset of pyrolysis but precedes ignition (t_M). The conditions to be studied are relevant to piloted ignition tests previously reported in the literature and thus provide a starting point for further investigations to build better chemistry and transport models to capture the polymer degradation process.

Piloted Ignition of PMMA

Piloted ignition tests were conducted with a 30 mm x 30 mm slab of 25 mm thick PMMA Type-G (Rohm and Haas). Tests were conducted by exposing the sample to a radiative heat flux originated from an electrical panel (Fig. 1). The pilot consists of an electrically heated coil of Kanthal wire placed above the sample. One surface was heated while the others were insulated.

The objective of these tests is to estimate the differences between the time to ignition (t_{ig}) and the time to attain the onset of pyrolysis (t_p), thus establishing the importance of the mixing time (t_M). For this purpose, type-K (0.5 mm) thermocouples are melted into the surface of the fuel slab. The relatively large bead for the thermocouples was to avoid breaking of the bead and minimize separation from the surface when the polymeric material loses its rigidity. Separation of the thermocouples from the surface limits the reliability of thermocouple measurements for establishing the onset of pyrolysis. Three other methods are used in conjunction with the thermocouple measurements: an infrared camera, mass loss measurements, and flow visualization. An Inframetrics-13463 infrared camera views the

surface of the fuel with a 3.399 μm filter that corresponds to the emissions of the monomer (MMA). The filter allows transmission of only 25% of the emission. The camera only records a slight temperature increase until the onset of degradation. The presence of the monomer results in a sudden increase in temperature induced by the contribution of the gas. This transition was used as an indicator for the onset of pyrolysis. The experiment was set on top of a load cell and the mass loss was obtained as a function of time with a Navigator N02120 scale that provided measurements with an accuracy of $\pm 0.01\text{g}$. The onset of pyrolysis was defined by a sudden increase in the mass loss rate. Finally, the onset of pyrolysis was determined by direct visualization of the smoke. The smoke was made visible by a light sheet originating from an SDL-7432 H1 Diode Laser that provides an output of 500 mW in a narrow band centered at 683 nm. The images were recorded with a COHU 4915 CCD camera with a 685 nm filter.

The time to the onset of pyrolysis (t_p) was recorded independently from all four different methods and tests were repeated at least five times for each heat flux. It was observed that the discrepancy between all methods was less than 10% therefore an average of all values will be presented here. Fig. 2 shows a characteristic surface temperature history. The experimental data shows a significant departure from the predicted solution using the inert heating of a semi-infinite solid for temperatures above 285°C. Beyond that point, the analytic solution over-predicts the temperature rise, which suggest an endothermic degradation process. The glass transition temperature for PMMA is $\sim 105^\circ\text{C}$ therefore a major change in thermo-physical properties being responsible for this departure was discarded. Ignition is represented by a sudden increase in the surface temperature ($t_{ig} \approx 1550\text{ s}$). It is important to note that Fig. 2 shows that $t_M (= t_{ig}-t_p)$ is slightly larger than t_p (850 s vs. 700 s).

The temperature for the onset of pyrolysis (285°C) remained consistent and was deemed as the characteristic pyrolysis temperature (T_p). t_p and t_{ig} were obtained for a wide range of heat fluxes, and the average results are presented in Fig. 3. As indicated in Fig. 3, there is a significant discrepancy between the two times, which indicated that the mixing time can not be neglected in ignition models. Furthermore, it can be easily shown that t_M is inherently linked to the flow characteristics [5], and thus deriving global thermal properties from the ignition delay times will result in values that are dependent on the environmental conditions of the test. This might be of little significance to the relative ranking of materials, as stipulated by the ASTM-E-

1321 standard [3] and later validated by Long et al. [1], but these discrepancies could be of relevance if these properties are used to model piloted ignition delay times.

Thermal Degradation of PMMA

The above observations show that adequate modeling of ignition delay times requires a proper estimation of the mixing time. Complete modeling of the mixing process that leads to formation of flammable mixtures requires quantification of both the rate and products of surface degradation chemistry as well as the different parameters that control the migration of gas phase degradation products through the molten material [11,12]. Although the work presented here does not determine these parameters and constants, it does establish the significance of transport (heat and mass) and chemistry in the degradation of PMMA.

Polymer degradation was measured with a TGA, (ThermoCahn model 2131). The products evolved were analyzed by a quadrupole MS (VG Gaslab300). The evolved gases from the PMMA samples were transferred through a capillary tube packed with fused silica. The transfer line was heated and maintained at temperature of 150°C to prevent condensation of water and heavier off-gases, flow through the TGA was fixed at 90 ml/min.

An industrial grade PMMA, commercial plexiglass type G (Rohm and Hass) was used for this study. The samples were obtained by grating larger pieces to particles of approximately 0.5 mm in diameter. The diameter was chosen based on the observation reported by Cordova et al. [5] that, even for heat fluxes above 40 kW/m², the thickness of the visible molten layer of a PMMA plate is greater than 1 mm. This choice for characteristic size is important since this will likely impact the influence of mass transport effects. Therefore, the characteristic size of the particle was set to be comparable to that of the degrading fuel in the ignition tests. The individual PMMA particles consist of chains of approximately 130,000 molecules and approximately 90 mg of material is spread over the surface of the container. The large sample size is necessary to obtain a good signal to noise ratio for the mass spectrometer.

For the determination of kinetic constants, it is critical to use smaller particles and samples, as reported elsewhere [13,14]. For this purpose, reference tests were conducted with powder PMMA that consisted of pure spheres of smaller diameter (approximately 0.1 mm) obtained from the Aldrich Company. The results will not be discussed here since they significantly departed from those obtained from the grated samples. The difference between

the chemically pure PMMA and the industrial samples can be attributed to the chemical additives of the industrial PMMA that included parting agents as well as ultraviolet absorbers. It is important to note that the thermo-grams for both industrial and powder PMMA also departed significantly from the results reported in references [13,14]. Since the variances in the chemical composition of all three Type-G PMMA samples could not be established, the study of these differences was considered beyond the scope of this work.

An initial set of TGA experiments was conducted with pure N₂ atmospheres. The normalized weight and the weight loss rates for different heating rates are presented in Fig. 4. It is important to note that the heating rates for ignition tests are never larger than 5°C/min. The weight loss rate curves with the smooth single peak, as shown in Fig. 4, suggest a one-step reaction for all heating rates. As the heating rate increases, the temperature at which maximum weight loss rate occurs increases, and the increase in the temperature of peak weight loss rate can be used to derive an effective global E_a (184 kJ/mol) for the degradation process.

More information of the degradation process can be gathered from the mass spectra curves from the real-time MS evolved gas analysis. Three molecular weight to charge ratios (m/z = 39, 41, and 55) are presented in Fig. 5. The m/z values in Figure 5 represent molecular fragments associated with MMA monomer fragmentation and thus the ion intensities curves for each m/z in Figure 5 show a similar shape and qualitatively represent the release of MMA monomer from the PMMA particles. For the low heating rate value of 1°C/min, the m/z curves show a clear peak, which correlates quite well with the peak in the weight loss rate. However, for the higher heating rate, the ion intensities for the key MMA fragments shows a rapid jump in evolved gases at 420°C followed by an equally rapid decay in the same ion intensities at 480°C. The rapid increase in MMA volatilization at the higher heating rates suggests one of two scenarios: 1) a high activation energy process beginning at approximately 420°C but mass transport limits the rate of reaction or 2) a rapid softening of the particles at 420°C that provides for rapid removal of trapped vapors inside the polymer particles. The mass transport limitation of the MMA production are likely due to inadequate transport of MMA vapors away from the polymer surface, since it is expected that the pure N₂ does not play a substantial role in the PMMA degradation kinetics. This apparent mass transport limitation was observed for all heating rates > 1°C/min. However, for 1°C/min, shown in Fig. 5, the rapid jump in reaction rates and apparent mass transport limitations are not observed and both the weight-loss and ion

intensity curves suggest that the process can be fairly well modeled by a one-step global mechanism. The impact of the heating rate on both the shape and peak location of the weight loss curves and the MMA vapor release suggests that the pyrolysis process is in some way self-promoting. This can be explained both by the reactions causing an increase in surface area as well as pyrolysis causing a local breaking of chains and thus an increase in exposed ends with enhanced activity for volatilization.

The presence of O_2 in the gas phase significantly reduces the temperature at which the degradation process is initiated. Experiments were conducted with 5%, 10%, 15% and 21% O_2 in N_2 . Fig. 6 shows how the initiation temperature for PMMA degradation, i.e. T_P (which is not necessarily T_{ig}), decreases by up to $100^\circ C$ with increasing O_2 content up to about 10% O_2 . Above 10% O_2 , increasing O_2 mole fraction does not seem to significantly lower T_P . Furthermore, the same sensitivity of T_P to heating rate observed in pure N_2 environments is observed for the O_2 containing environments. This variation in heating rate is extremely significant for modeling piloted ignition because it suggests that temperature is not the only factor governing the pyrolysis and degradation processes responsible for piloted ignition. Typical degradation temperatures for PMMA have been reported to be $> 300^\circ C$ [13,14] and it is not surprising that these values have been unsuccessful in predicting piloted ignition. Heating rate and associated transport effects that can only be established when analyzing the degradation products might lead to a bias for these values.

The characteristic pyrolysis temperatures established during piloted ignition tests ($285^\circ C$) fall somewhere in between the two temperature ranges. This suggests that a combination of inert and oxidative pyrolysis might provide the gaseous fuel necessary for ignition. This merits a closer observation of the degradation of PMMA in an oxidative atmosphere. Fig. 7 shows evolution of the weight loss rate as a function of temperature for 5% and 21% O_2 in N_2 – in each case for two different heating rates. The weight loss rate curves show that the onset of degradation in the oxygenated environments occurs at approximately $250^\circ C$. As the temperature increases, the weight loss rate increases until a sudden increase in the rate is observed at $T \approx 360^\circ C$. This initial step in the degradation process has been previously described by Hirata et al. [14]. Degradation follows with a weaker increase in the weight loss rate. The first rate change can be attributed to the limited amount of oxygen that can diffuse into the material and that is necessary for the oxidative pyrolysis. Nevertheless, this

process seems to contribute to the gasification of the material since the weight loss rate continues to increase. A second rapid increase in weight loss rate is observed at $T \approx 360^\circ\text{C}$, which is clearly associated with a second step in the polymer degradation. The cause for this second step is uncertain, but it is clear that oxidative attack of the surface influences this process because a similar peak was observed at much higher temperatures $T \approx 420^\circ\text{C}$ in the pure N_2 environments. The intensity of this second peak seems to increase with O_2 concentration and decrease with heating rates. Thus, O_2 transport to the surface must be limiting the degradation reactions for $T < 360^\circ\text{C}$ and the increase in weight loss rate might be due to an increase in diffusivity resulting from the temperature increase. This is important since piloted ignition temperatures for conditions similar to those of ASTM-E-1321 are generally lower than 360°C (Fig. 2).

A plateau that is more evident for the lower heating rates follows the second inflection point. This phenomenon seems to contradict the conclusion that at this stage mass transport of oxygen is the limiting parameter of degradation, but might be the result of two competing mechanism, in-depth inert degradation and surface oxidation. As the temperature increases, inert degradation becomes dominant and a second sudden increase in the weight loss rate is observed between $430\text{-}450^\circ\text{C}$. These temperatures correspond well with those presented in Fig. 4 and 5.

Figure 8 shows some characteristic m/z ratios for air. The conditions are the same as the mass loss rate curves presented in Fig. 7. The m/z ratios provide little insight on the initiation of oxidative pyrolysis since the MS sensitivity was unable to pick up the volatilization products due to the low production rates for $T < 300^\circ\text{C}$. The evolution of the m/z ratios for $3^\circ\text{C}/\text{min}$ confirms conclusions derived from Fig. 7. Furthermore, it shows that the final peak is actually truncated as during inert degradation. It is of extreme importance to point that both steps of the degradation are impacted by transport. Mass transport in and out of the fuel is a phenomenon that is not well understood [12] but all too often neglected. The $m/z = 44$ curve – representative of CO_2 , is presented in Fig. 8 is not noticed in inert degradation and thus shows that the presence of O_2 is playing a significant role in the products of the degradation and thus in the formation of a potentially flammable mixture. Thus, the importance of environmental O_2 concentration even for piloted ignition studies is critical since O_2 will impact not only the

pyrolysis initiation temperature but also the degradation gases that will go into the formation of the flammable gas-phase mixture.

Conclusions

Degradation of Type-G PMMA (Rohm and Haas) in pure N₂ and oxygenated atmospheres was studied. Tests conducted with a PMMA slab heated by a radiant panel showed that:

- Degradation prior to piloted ignition can account for more than 50% of the ignition delay time.

The degradation of the polymer was measured with a TGA and a MS analyzed the products evolved. It can be concluded that:

- Mass transport limitations sometimes not observable through mass loss curves are indicated as abrupt changes in the evolved gas profiles. This indicates the importance of coupling TGA measurements with evolved gas analysis for detecting mass transport limitations.
- Mass loss measurements suggested that degradation of industrial PMMA in pure N₂ can be modeled as a single-step reaction, but further analysis of the evolved gases showed that mass transport effects affected the degradation process.
- The presence of O₂ shifted the degradation to lower temperatures. The analysis of the evolved gases suggested a two-step reaction and again revealed mass transport effects. Oxygen transport to the surface limits the degradation reactions for $T < 350^{\circ}\text{C}$. For $350^{\circ}\text{C} < T < 430^{\circ}\text{C}$, inert and oxygen promoted degradation compete with inert degradation becoming dominant for $T > 430^{\circ}\text{C}$.

The present results have shown that, at least for PMMA, modeling of piloted ignition requires not only an appropriate definition of the kinetic constants but also adequate modeling of oxygen transport into the material and gasified combustibles out of the sample.

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FIGURE CAPTIONS

Figure 1 – Schematic of the experimental setup used to obtain the ignition delay time (t_{ig}) and the time for the onset of pyrolysis (t_p).

Figure 2 – Characteristic surface temperature history corresponding to an experiment conducted with $\dot{q}_e'' = 12 \text{ kW/m}^2$. Two curves are shown, one corresponding to the semi-infinite solid solution under the assumptions given by Quintiere [2] and with global properties as defined by reference [5].

Figure 3 – Characteristic ignition delay times (t_{ig}) and times to the onset of pyrolysis (t_p) for PMMA and a wide range of external heat fluxes.

Figure 4 – Weight loss and weight loss rate curves for different heating rates in pure N_2 for 0.5 mm samples of industrial-grade PMMA.

Figure 5 – Weight loss rate and molecular weight to charge ratios for industrial-grade PMMA degradation in a pure N_2 atmosphere. The thinner lines correspond to $5^\circ\text{C}/\text{min}$ heating rates while the thicker lines to $1^\circ\text{C}/\text{min}$.

Figure 6 – Onset of degradation for different oxygen concentrations and heating rates. The intercept between the horizontal (initial mass of the sample) and a tangent to the weight loss curve at the maximum weight loss rate temperature defines the onset of degradation. The outlying data point for 5% oxygen is probably due to moisture in the sample.

Figure 7 – Weight loss rate curves for air (thick line) and 5% O_2 degradation (thin line). Two heating rates are presented for each oxygen concentration.

Figure 8 – Weight loss rate and molecular weight to charge ratios for PMMA degradation in an oxygenated atmosphere. The thin lines correspond to $3^\circ\text{C}/\text{min}$ heating rates while the thick lines to $5^\circ\text{C}/\text{min}$.

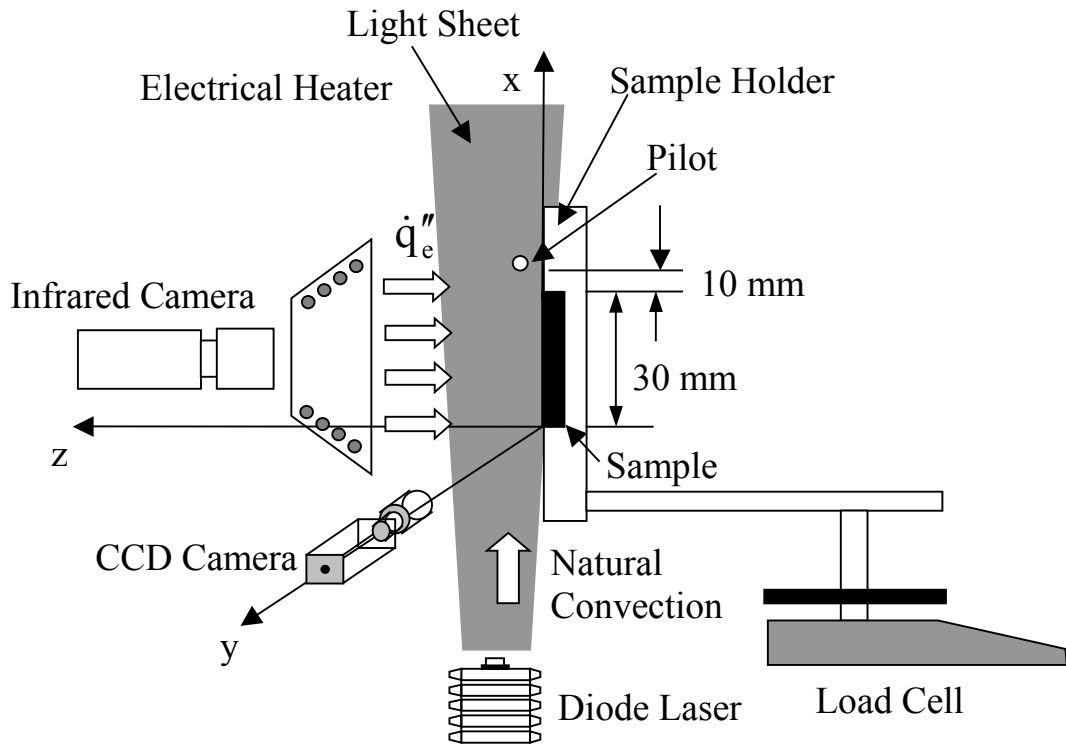


Figure 1

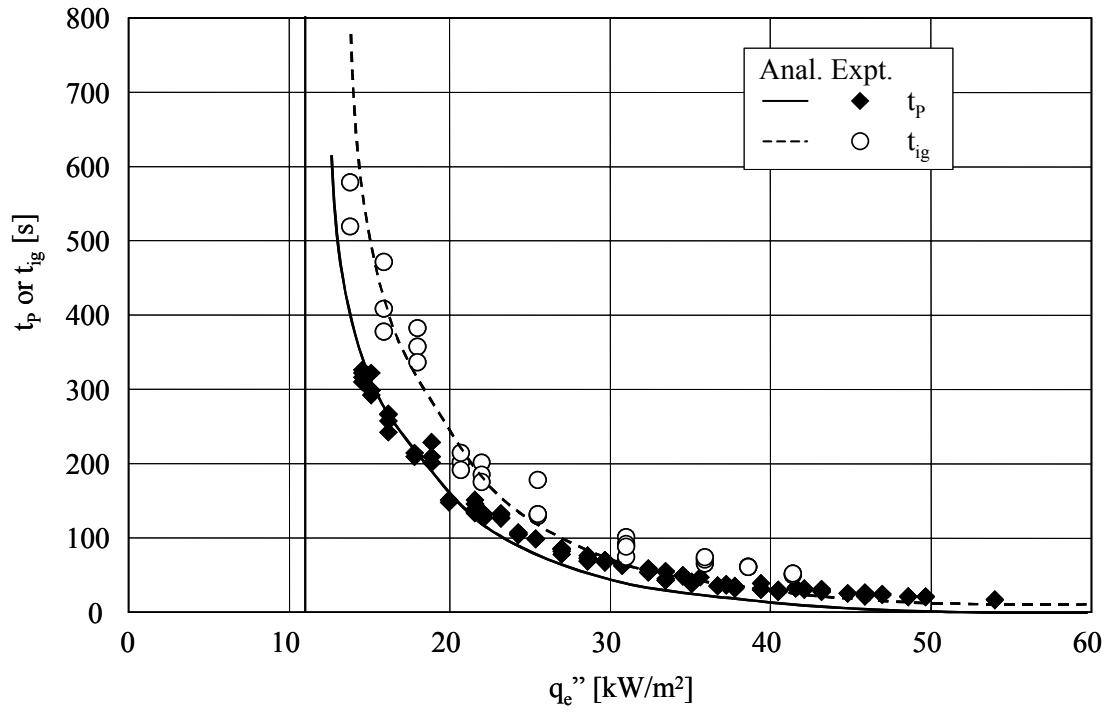


Figure 3

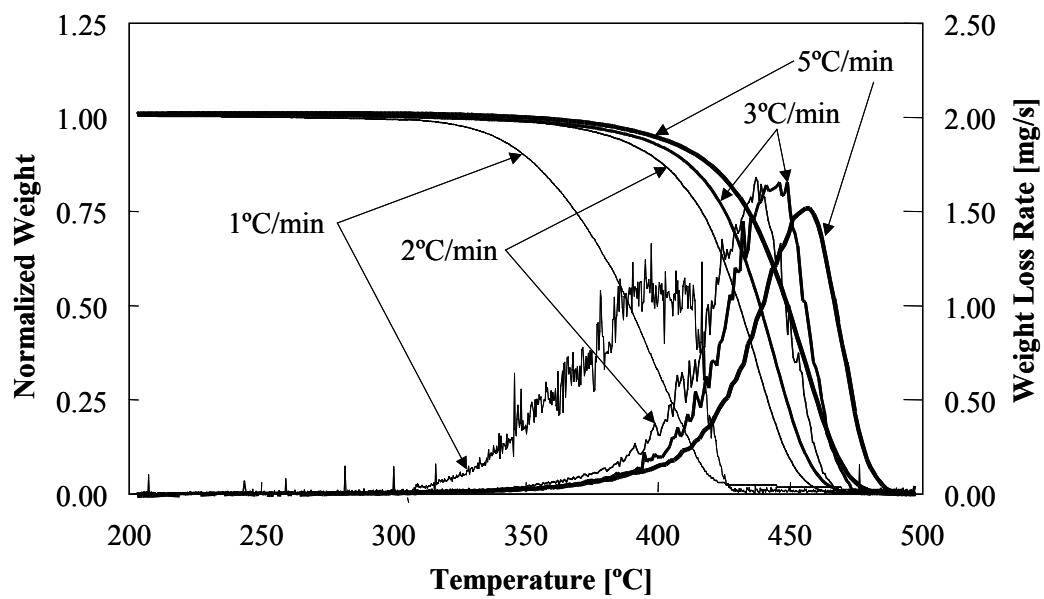


Figure 4

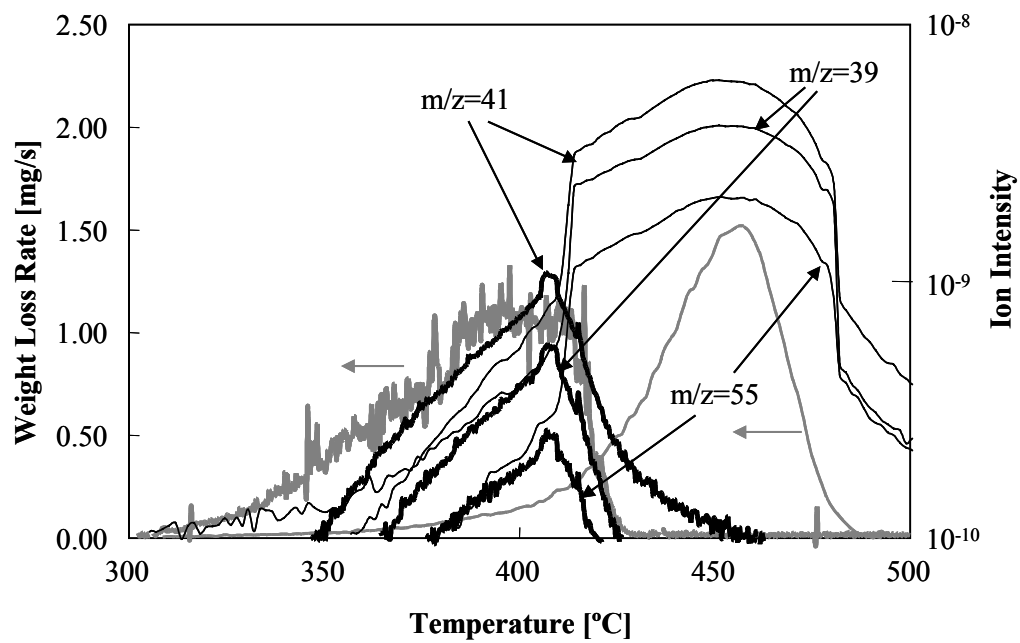


Figure 5

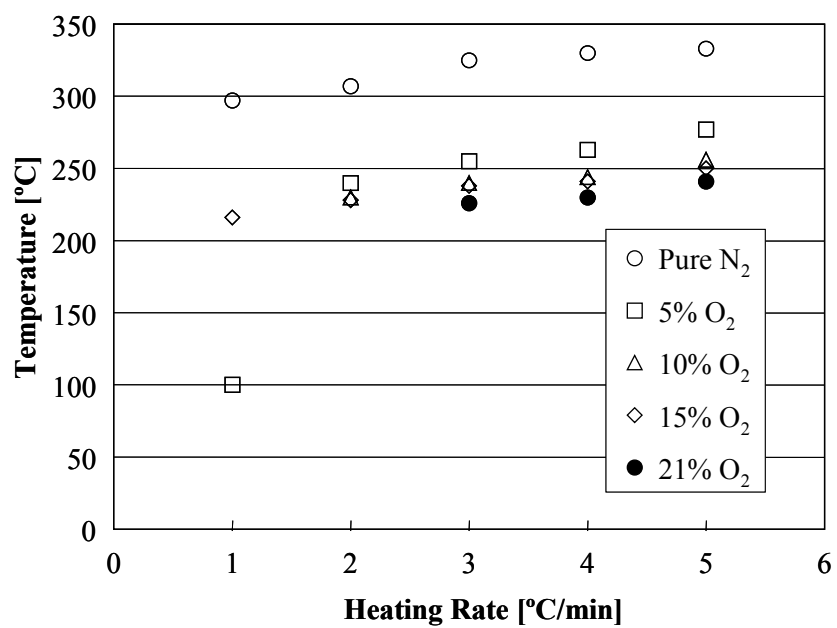


Figure 6

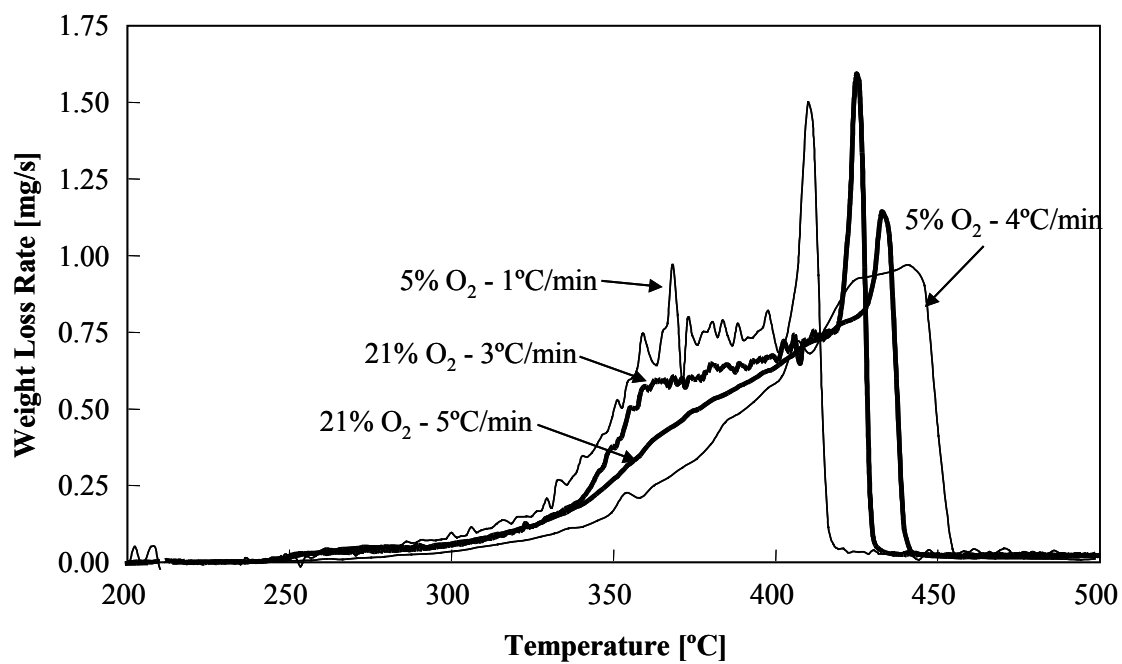


Figure 7

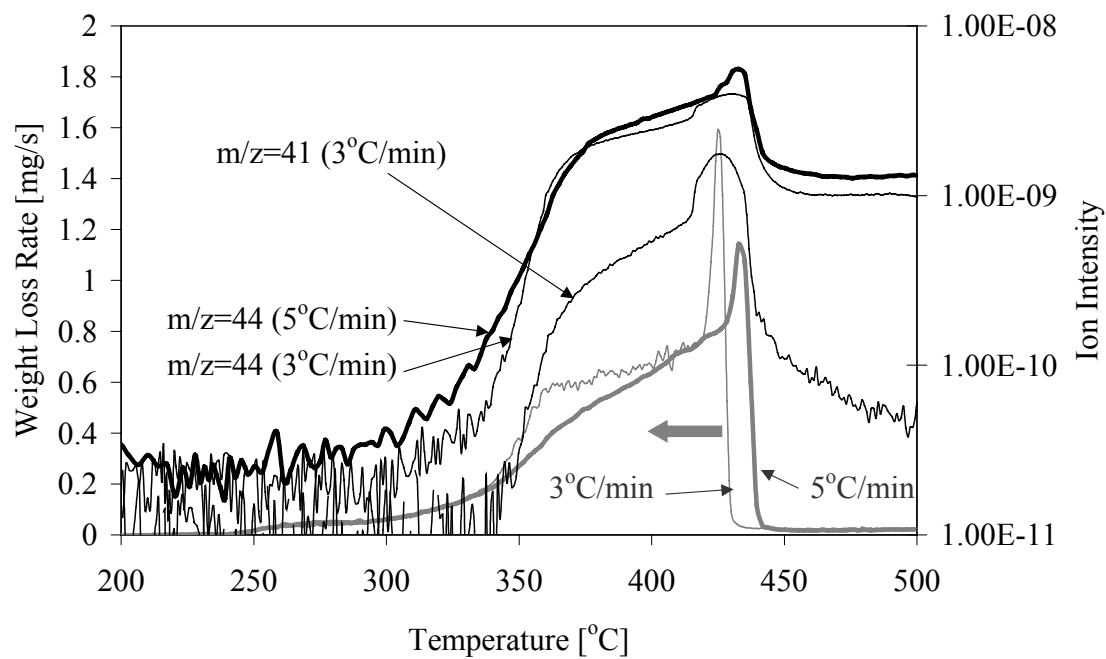


Figure 8